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Low efficiency roll-off blue TADF OLEDs employing a novel acridine–pyrimidine based high triplet energy host†

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The development of efficient blue emitter–host combinations is one of the biggest challenges in organic light-emitting diode (OLED) research. Host materials play a crucial role when it comes to enhancing the efficiency, improving the lifetime and decreasing the efficiency roll-off of the device. The need for new hosts is of prime importance, especially for blue phosphorescence and thermally activated delayed fluorescence (TADF) emitters, due to their high exciton energies. The hosts are less investigated than the emitters and require further progress. This work provides a new molecular strategy that combines an acridine derivative (donor) and pyrimidine moieties (acceptors) to obtain three novel host materials. This approach demonstrates that *via* careful selection of donor and acceptor units, it is possible to manage the properties of the host materials, obtaining at the same time superior thermal and morphological properties and high triplet energies up to 3.07 eV. The decrease of the conjugation in the acceptor unit was found to play a crucial role in increasing the triplet energy. The most promising host **1MPA** was used to fabricate blue TADF OLEDs. Using a sky-blue emitter, we achieved electroluminescence at 491 nm and a maximum external quantum efficiency (EQE) of 13.6%, combined with a low efficiency roll-off, even beyond the practical brightness of 1000 cd m⁻¹. The host **1MPA** was also combined with a deep blue emitter to deliver a blue OLED with color coordinates of $x = 0.16$ and $y = 0.18$.

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Introduction

Presently, organic light-emitting diodes (OLEDs) are expanding their boundaries and applications, such as in flat panel displays and smartphones, due to their significant advantages like facile preparation, cost-effectiveness, and energy efficiency.^{1,2} For these reasons, much effort is focused on developing this technology, both in academia and industry. Especially for display applications, commercial OLEDs contain stable and efficient green and red emitters based on iridium complexes.³ In contrast, blue emitters suffer from lacking simultaneous efficiency and

stability.^{4,5} In the last few years, thermally activated delayed fluorescence (TADF) emitters seemed to be among the most promising candidates to solve this problem. Their capacity of harvesting triplet excited states can allow for an internal quantum efficiency of up to 100% through prompt and delayed fluorescence emission. Rapid thermally activated transfer of the excitation from the triplet to the singlet state, followed by efficient delayed fluorescence, avoids degradation processes that could otherwise occur when the excitation prevails for a long time in the triplet state. TADF OLEDs contain purely organic molecules, which avoid rare and costly noble metals and allow a broad range of molecular engineering. Aside from emitters, host materials play an essential role in obtaining the desired performance of the OLED. Emitters are frequently doped in a host matrix to decrease detrimental bimolecular effects caused by longer-lived triplet states.⁶ The dilution into the host is thus essential to enhance the lifetime, increase the efficiency, and reduce the efficiency roll-off of the device.⁷ The hosts, in general, must have high and balanced charge carrier mobility, good thermal and morphological stability, and chemical stability. They also must have high triplet energy to confine the excitons on the emitter, which is particularly challenging for blue-emitting OLEDs.⁸

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The majority of host materials adopt carbazole or less stable phosphine oxide groups.⁹ In this work, we present host molecules based on acridine and pyrimidine groups instead. Acridine is often used as a donor (D) in emitter molecules because of its rigidity and strong donating feature.^{10–13} Few cases report its use in hosts for red phosphorescence OLEDs.^{14,15} Furthermore, what brought our attention to the acridine group is that it has a higher triplet energy than carbazole (due to less conjugation).¹⁶ The pyrimidine unit is also a building block for TADF emitters due to its acceptor (A) properties.^{17–20} It is also used as a building block for hosts in phosphorescence OLEDs, with triplet energies of these hosts lying below 3 eV.^{21–23} Another use of pyrimidine is as an electron transport material (ETM).²⁴

In this work, we report a strategy to obtain high triplet energy donor–acceptor host materials based on acridine and pyrimidine moieties. In particular, *via* careful molecular modification, it is possible to limit the conjugation of the acceptor pyrimidine, leading to a high triplet energy of more than 3 eV. To prove the strategy's applicability, we fabricated blue TADF OLEDs using the most promising host of the series (**1MPA**), expressing blue emission (from 461 nm to 493 nm) with low efficiency roll-off even beyond a practical brightness of 1000 cd m⁻².

Experimental

Materials

9,9-Diphenyl-9,10-dihydroacridine; 2-chloro-4,6-diphenylpyrimidine; 4,6-dichloro-2-phenylpyrimidine; 4,6-dichloro-2-methylpyrimidine; palladium(II) acetate and tri(*tert*-butyl)phosphine (10 wt% in hexane) were purchased from abcr. Sodium *tert*-butoxide was purchased from TCI.

Synthesis

10-(4,6-Diphenylpyrimidin-2-yl)-9,9-diphenyl-9,10-dihydroacridine (**2PPA**). 9,9-Diphenyl-9,10-dihydroacridine (1 g, 3 mmol, 1.1 eq.), 2-chloro-4,6-diphenylpyrimidine (727 mg, 2.73 mmol, 1 eq.), sodium *tert*-butoxide (340 mg, 3.54 mmol, 1.3 eq.), palladium(II) acetate (25 mg, 0.11 mmol, 0.04 eq.) and toluene (25 ml) were placed with a stirring bar in a Schlenk tube. The mixture was then degassed three times through freeze–pump–thaw cycles and the tube was backfilled with argon. Subsequently, tri(*tert*-butyl)phosphine(10 wt% in hexane) (0.07 ml, 0.22 mmol, 0.08 eq.) was added and the mixture was then refluxed for 12 h. The reaction mixture was extracted with water and the organic phase was dried over anhydrous sodium sulphate before evaporating the solvent with reduced pressure. Finally, the compound was purified by column chromatography on silica gel (20% ethyl acetate/hexane) and by train sublimation, obtaining 25% yield of pure product. ¹H NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 8.01 (d, *J* = 7.5 Hz, 2H), 7.92 (d, *J* = 7.5 Hz, 4H), 7.45–7.42 (m, 7H), 7.39 (t, *J* = 7.5 Hz, 2H), 7.17 (t, *J* = 7.5 Hz, 2H), 7.05–6.99 (m, 6H), 6.96 (d, *J* = 7.0 Hz, 2H), 6.90 (d, *J* = 6.5 Hz, 4H). ¹³C NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 164.21, 160.05, 144.35, 142.55,

140.91, 137.73, 130.63, 130.27, 128.60, 128.04, 127.45, 127.28, 127.04, 126.32, 125.55, 124.07, 104.45, 58.62. EI-MS *m/z* [M]⁺: 564.

10,10'-(2-Phenylpyrimidine-4,6-diyl)bis(9,9-diphenyl-9,10-dihydroacridine) (**1PPA**). **1PPA** was synthesized following a similar procedure as described for **2PPA**; using 9,9-Diphenyl-9,10-dihydroacridine (977 mg, 2.93 mmol, 2.2 eq.), 4,6-dichloro-2-phenylpyrimidine (300 mg, 1.33 mmol, 1 eq.), sodium *tert*-butoxide (320 mg, 3.33 mmol, 2.5 eq.), palladium(II) acetate (25 mg, 0.11 mmol, 0.08 eq.), tri(*tert*-butyl)phosphine (10 wt% in hexane) (0.07 ml, 0.22 mmol, 0.16 eq.) and toluene (25 ml). Yield = 28%. ¹H NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 8.27 (m, 2H), 7.44–7.37 (m, 7H), 7.22 (t, *J* = 7.5 Hz, 4H), 7.16–7.10 (m, 12H), 7.07 (t, *J* = 7.5 Hz, 4H), 6.89 (d, *J* = 7.5 Hz, 4H), 6.82 (d, *J* = 7.5 Hz, 8H), 5.65 (s, 1H). ¹³C NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 162.68, 161.44, 144.60, 140.17, 140.15, 138.37, 130.40, 130.15, 128.93, 128.30, 128.17, 127.56, 126.63, 126.09, 123.74, 123.69, 91.1, 58.12. EI-MS *m/z* [M]⁺: 818.

10,10'-(2-Methylpyrimidine-4,6-diyl)bis(9,9-diphenyl-9,10-dihydroacridine) (**1MPA**). **1MPA** was synthesized following a similar procedure as described for **2PPA**; using 9,9-diphenyl-9,10-dihydroacridine (2.35 g, 7.05 mmol, 2.3 eq.), 4,6-dichloro-2-methylpyrimidine (0.50 g, 3.07 mmol, 1 eq.), sodium *tert*-butoxide (0.77 g, 8.01 mmol, 2.6 eq.), palladium(II) acetate (55 mg, 0.25 mmol, 0.08 eq.), tri(*tert*-butyl)phosphine(10 wt% in hexane) (0.15 ml, 0.49 mmol, 0.16 eq.) and toluene (25 ml). Yield = 58%. ¹H NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 7.23 (d, *J* = 7.5 Hz, 4H), 7.18 (t, *J* = 7.5 Hz, 4H), 7.14–7.10 (m, 12H), 7.04 (t, *J* = 7.5 Hz, 4H), 6.87 (d, *J* = 7.5 Hz, 4H), 6.80 (d, *J* = 7.5 Hz, 8H), 5.60 (s, 1H), 2.43 (s, 3H). ¹³C NMR (CDCl₃, 500 MHz, Me₄Si): δ [ppm] 144.76, 140.07, 138.81, 130.34, 130.06, 129.01, 127.88, 127.55, 126.60, 126.28, 123.36, 122.31, 94.77, 57.92, 26.17. EI-MS *m/z* [M]⁺: 756.

Purification

The synthesized compounds **2PPA**, **1PPA**, and **1MPA** were purified by train sublimation in a Carbolite split tube furnace HZS 12/450.

Characterization

¹H NMR and ¹³C NMR spectra were recorded on a Bruker Avance III HD (500 MHz), the chemical shifts were referred to chloroform-d₃ (7.26 ppm), and the *J* values are given in Hz. MS spectra were obtained on a Finningan MAT 8500 using electron impact ionization.

Thermal measurements

Thermogravimetric analysis (TGA) was conducted on a Mettler TGA/DSC3 with a heating rate of 10 K min⁻¹ under a nitrogen flow. Differential scanning calorimetry was performed with a Mettler DSC3+ in pierced Al pans at 10 K min⁻¹ under a nitrogen flow. The glass transition *T*_g was determined as the midpoint temperature of the step. *T*_m and *T*_r were determined as the peak temperature of the melting peak and recrystallization peak, respectively. These calculations were performed by Mettler STARE 15.00a software.



Electrochemical measurements

Cyclic voltammetry measurements were carried out using a standard three electrode electrochemical micro-cell kit from Ametek Scientific Instruments, consisting of a Platinum wire counter electrode, a Platinum disk working electrode, and a Silver/Silver chloride reference electrode. A Gamry Interface 1010T served as the potentiostat. Electrochemistry grade solvent was used, and 100 mM tetrabutylammonium hexafluorophosphate was the supporting electrolyte. The solutions were thoroughly purged with dry nitrogen for 10 minutes before each measurement.

Computational details

We have used density functional theory to obtain the ground state geometries of the compounds employing the M06-2X exchange–correlation functional in combination with a 6-31+G(d,p) atomic basis set. Geometry optimizations were successfully converged, and no imaginary frequencies were observed following vibrational frequency analysis. Singlet and triplet emission energies were further obtained by optimizing the structures with linear response time-dependent density functional theory at the M06-2X/6-31+G(d,p) level²⁵ and within the Tamm–Dancoff approximation.²⁶ Electron–hole natural transition orbitals (NTOs) were calculated to characterize the excited state transitions.²⁷ All calculations were performed using the Gaussian 16 software package,²⁸ and orbitals were visualized using the Avogadro molecular editor software.²⁹ Charge transfer numbers ω_{CT} with values in the range between 0 (for a Frenkel exciton) and 1 (for a complete charge transfer exciton) were computed by a transition density matrix analysis,^{30,31} based on defragmentation of the molecules in acridine and pyrimidine groups (D–A–D for **1PPA** and **1MPA** and D–A for **2PPA**). Electron–hole correlation plots of the Ω matrix for the excitations were also constructed based on the fragments. The average electron–hole distance Δr , based on the charge centroids of the orbitals involved in the transition,³² was also computed as a complementary indicator to the nature of the states using Multiwfn software.³³

Photophysical measurements

For measurements in solution, all the compounds were dissolved in toluene or mTHF at a concentration of $6\text{--}9 \times 10^{-5}$ M. Solutions were sonicated for 15 minutes after preparation. Absorption spectra were recorded using a Varian Cary 5000 spectrophotometer. Fluorescence spectra, phosphorescence spectra, and room temperature (RT) photoluminescence quantum yield (PLQY) were measured using a Jasco FP-8600 spectrofluorometer. For phosphorescence measurements, the detector unit was opened at a delay of 150 ms after excitation, and the signal was acquired for 50 ms. The excitation wavelength for both fluorescence and phosphorescence was 300 nm (unless specified in the text or figure). For the 77 K measurements, the sample or cuvette was immersed in liquid nitrogen. Values for the photoluminescence quantum yield of the molecules in solution were obtained using the Jasco FP-8600 spectrofluorometer equipped with an integrating sphere.

OLED fabrication and characterization

OLEDs were fabricated on pre-patterned indium tin oxide (ITO)-coated glass substrates (Kintec) with an ITO layer thickness of 100 nm and a sheet resistance of $15\text{--}20 \Omega \square^{-1}$. Prior to device fabrication, the substrates were cleaned by sonicating consecutively in detergent (Hellmanex II), distilled water, acetone, isopropyl alcohol, boiling distilled water, and finally treated with O_2 -plasma for 10 min. The substrates were then transferred into a vacuum evaporation chamber (Vacuum Systems and Technologies Ltd) integrated inside the nitrogen-filled glove box, where the stack of organic layers was deposited at a rate of $0.5\text{--}1 \text{ \AA s}^{-1}$ and a pressure $<10^{-6}$ Torr. The doping of the emissive layer was accomplished by co-evaporating the host and guest materials from different evaporation sources at distinct rates for the desired doping concentration. Afterward, the samples were transported by the robotic arm to a metal deposition chamber without breaking the vacuum, which was followed by deposition of lithium fluoride (LiF) and aluminum (Al) layers at a rate of 0.2 and 2 \AA s^{-1} , respectively. The active area of the devices was 4 mm^2 as defined by the ITO pattern and the shadow mask used for cathode deposition. The completed devices were removed from the vacuum chamber to the N_2 -filled glove box without exposure to air and encapsulated by using a glass cover and UV-curable epoxy KATIOBOND LP655 (DELO). A system consisting of a calibrated integrating sphere (ORB Optronix), a spectrometer PMA-11 (Hamamatsu), and a source-meter unit 2601A (Keithley) was utilized for evaluation of the electrical-optical properties of OLEDs such as current–voltage–luminance (I–V–L) characteristics, EQE and efficiency roll-off.

Results and discussion

Design and synthesis

In this work, a strategy to increase the triplet energy of donor–acceptor molecules is shown. For this purpose, three novel host molecules were designed. As presented in previous work,³⁴ by direct connection of donors (acridine and carbazole) and acceptors (triazine), it is possible to increase the triplet energy of donor–acceptor molecules compared to the case where the connection occurs *via* a conjugated bridge. In this work, acridine (donor) and three different pyrimidines (acceptors) with a different size of the conjugated system of the pyrimidine were chosen to obtain three materials of different conjugation. In particular, we aim to see a decrease in the conjugation of the acceptor moiety, a key strategy to reach a high-energy triplet (Fig. 1). **2PPA** presents a donor–acceptor structure where the pyrimidine core is linked at positions 4 and 6 to two phenyl rings and at position 2 to one acridine. **1PPA** has a donor–acceptor–donor (DAD) structure; *i.e.*, the pyrimidine moiety is linked to just one phenyl ring (in position 2) while two acridines occupy positions 4 and 6. The last molecule of the series, **1MPA**, still has the same DAD configuration, but the pyrimidine is substituted with a methyl group (in position 2) and two acridines (in positions 4 and 6). By replacing the phenyl ring with a methyl group, the conjugation of the



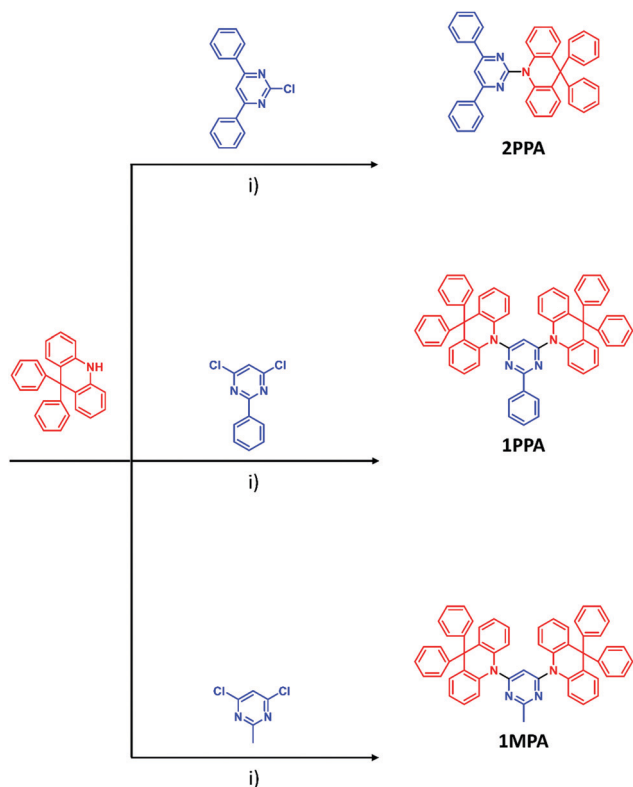


Fig. 1 Synthesis route to the three host materials **2PPA**, **1PPA** and **1MPA**. (i) Pd(OAc)₂, P(tBu)₃, NaOtBu, toluene, reflux, 12 h.

acceptor is limited to the central pyrimidine ring. The products were synthesized using a general Buchwald-Hartwig protocol between the donor acridine and the chlorinated pyrimidine acceptor, which provides an efficient way for the formation of C–N bonds (Fig. 1). Furthermore, all products were purified *via* train sublimation to obtain highly pure compounds.

Thermal and electrochemical analysis

The thermal properties of the compounds were investigated by differential scanning calorimetry (DSC) and thermogravimetric analysis (TGA), as shown in Fig. 2a and b. DSC shows that all the compounds are morphologically stable beyond 100 °C with glass transitions (T_g) of 108 °C (**2PPA**), 150 °C (**1PPA**), and 138 °C (**1MPA**) and that recrystallization (T_r) upon heating occurs above 100 °C for **2PPA** and above 200 °C for **1PPA** and **1MPA**, due to the addition of a second bulky acridine. The heating scan of **2PPA** was obtained after the sample was melted and directly cooled with liquid nitrogen. This was necessary to obtain the amorphous material during the cooling process. The melting temperatures (T_m) are around 300 °C for **2PPA** and **1PPA** and 273 °C for **1MPA**.

TGA data show high decomposition temperatures (T_d) above 300 °C for **2PPA** and above 400 °C for **1PPA** and **1MPA**. TGA also indicates that all the materials leave none or only minor residues which means that all three hosts completely evaporate already at normal pressure, which makes them ideal candidates for OLED fabrication *via* evaporation. Furthermore, throughout

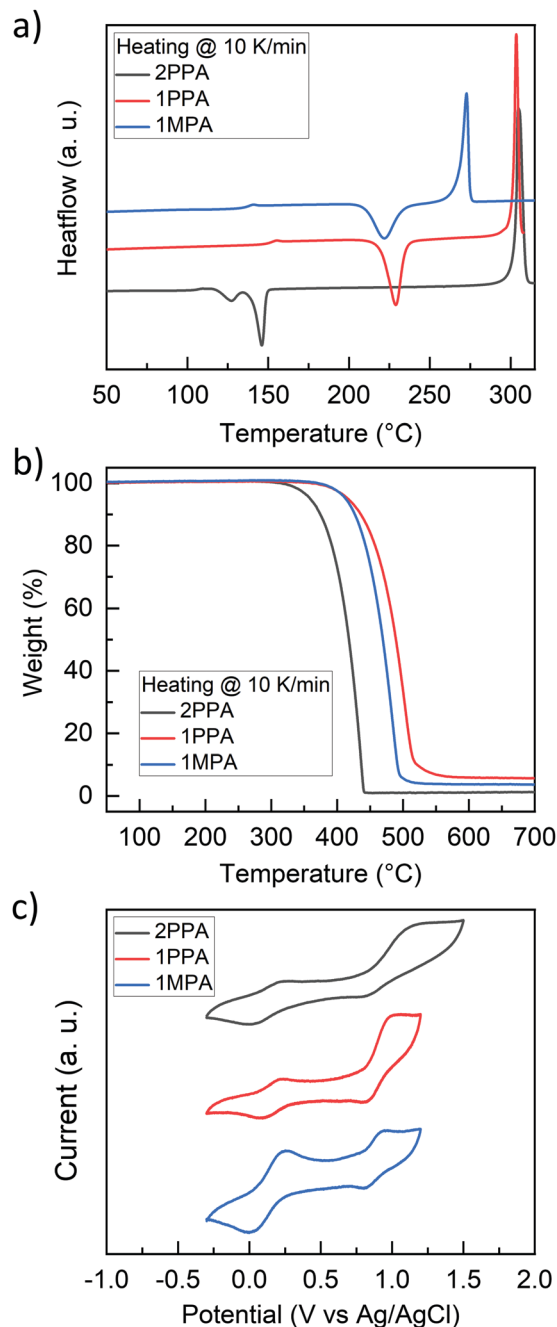


Fig. 2 Thermal and electrochemical properties of **2PPA**, **1PPA** and **1MPA**. (a) Differential scanning calorimetry showing the second heating scan. (b) Thermogravimetric analysis. (c) Cyclic voltammograms (cathodic scan) measured in CH₂Cl₂ at room temperature with tetra-*n*-butylammonium hexafluorophosphate (0.1 M) as the supporting electrolyte and a scan rate of 100 mV s⁻¹.

the series, it is possible to see that by adding a second acridine, the T_g , T_r , and T_d values of the materials are improved. The melting points are instead similar, with a small decrease in **1MPA**. These data confirm the high thermal stability of the materials and their potential applicability as hosts in OLEDs.

Furthermore, cyclic voltammetry of the three hosts was carried out to explore the electrochemical properties and to estimate the HOMO values (Fig. 2c). All the compounds show



two quasi-reversible oxidation peaks. The first oxidation peak always appears slightly above 0.2 V vs. Ag/AgCl, and it corresponds to the oxidation of the acridine donor moiety. The HOMO values for the three compounds were estimated by the half-wave potential of the first oxidation potential, considering the solvent correction according to Gräf *et al.*³⁵ The LUMO values were then calculated by adding the optical band gap to the HOMO values (Table 1).

Absorption and photoluminescence in solution

Absorption and photoluminescence studies of **2PPA**, **1PPA**, and **1MPA** were performed in dilute toluene solutions to analyze their intrinsic photophysical properties, with a focus on determining their triplet state energies. Fig. 3a shows the room temperature absorption spectrum of **2PPA**, **1PPA**, and **1MPA**. **2PPA** exhibits a band attributed to CT absorption from around 3.2–3.9 eV. This assignment is based on the structureless absorption profile and comparatively low extinction of this particular band. **1PPA** and **1MPA** do not show any pronounced signatures of a CT state absorption. The absorption of **2PPA**, **1PPA**, and **1MPA** differ significantly from the absorption of either donor or acceptor (Fig. S1a–c, ESI†). This indicates the existence of strong electronic interactions between the donor and acceptor units in these compounds.

In order to estimate the triplet and singlet energies of the host materials, phosphorescence (Ph) and steady-state (SS) measurements were carried out at liquid nitrogen temperature (77 K). For **2PPA** (Fig. 3b), the phosphorescence is structured with a 0–0 vibrational peak at 2.81 eV (442 nm). The steady-state spectrum is a combination of fluorescence and phosphorescence as the spectrum below 2.9 eV matches completely with the phosphorescence spectrum. Thus, the lowest singlet in **2PPA** is a state exhibiting a structureless emission centered around 3 eV, with an onset at 3.15 eV (394 nm), with the lack of vibrational structure probably indicating a CT-character. In **1PPA**, the lowest singlet and triplet states are both structured and have 0–0 peaks at 3.36 eV (369 nm) and 2.76 eV (449 nm), respectively, as shown in Fig. 3c. Thus, the T_1 state is at a lower energy in **1PPA** than in **2PPA**. This can be accounted for by the nature of this state. The phosphorescence of **2PPA** and **1PPA** are similar to the ones of their respective pyrimidine acceptors (Fig. S2, ESI†), indicating that the triplet is localized on the pyrimidine acceptor moieties, 2PhPy and 1PhPy, in agreement with the quantum chemical calculations (*vide infra*). The triplet energy of the hosts therefore reflects the phosphorescence energies of the respective

pyrimidine acceptors. The lower T_1 energy in **1PPA** results from the fact the phenyl ring is planar with the pyrimidine in **1PPA** whereas there is a torsion in **2PPA**, as evident in the quantum chemical calculations further below.

Due to the strong emission from the pyrimidine-based singlet state in **1PPA**, we cannot discern whether there is also a CT singlet state at about 3 eV with a lower oscillator strength buried under it. When we measured the spectra with different excitation wavelengths between 300 nm and 350 nm, we found them to be independent of excitation wavelength (Fig. S3, ESI†).

The emission of **1MPA**, in contrast, depends on the excitation wavelength, and the corresponding spectra are shown in Fig. 3d. As before, Ph and SS measurements were carried out at 77 K. For excitation at 300 nm and 320 nm (4.13 eV and 3.88 eV, respectively), we obtain a structured phosphorescence band with a 0–0 peak at 3.18 eV. However, when exciting at 335 nm and at 350 nm (3.70 eV and 3.54 eV), the emission band has a similar shape, yet it is shifted to lower energy, and it is broadened with a shoulder at 2.88 eV and an onset at 3.07 eV. To clarify the origin of these two bands, we performed photoluminescence excitation (PLE) spectra, and we also measured the phosphorescence of the acridine donor as well as the methylpyrimidine acceptor. These data are available in the ESI† (Fig. S4 and S5). For the higher energy phosphorescence, two assignments are principally possible, that is, to a locally excited (LE) state localized either on the acridine donor or on the pyrimidine acceptor. Based on the data in Fig. S4 and S5 and the arguments detailed in the ESI†, we tentatively assign the higher energy phosphorescence band to a 3LE state localized on the acridine donor, and we attribute the lower energy phosphorescence band to a triplet state with mixed CT–LE character, where the LE contribution would be again from the acridine.

The fluorescence in **1MPA** shows the same excitation dependence. For excitation at 300 nm and 320 nm, we observe a broad band centered at about 3.6 eV (Fig. 3e), while for excitation at 335 nm and 350 nm, the fluorescence is centered at 3.0 eV (Fig. 3f). By comparison to the acridine fluorescence spectrum (Fig. S4, ESI†), the band at 3.6 eV is readily assigned to a 1LE state localized mainly on the acridine. The band at 3.0 eV, by virtue of its lower energy, must have some significant CT contribution.

We also determined the S_1 and T_1 energies of all compounds in neat films at 77 K and found them to be shifted to lower energies by 0.1 eV (see Fig. S6, ESI†). We attribute this rather moderate shift in the neat film to energy transfer within the DOS, which is known to be efficient in neat films, so that the emission comes from sites with lower energy within the DOS.

Fig. 4a summarizes the insights obtained on the energies and character of the excited states in **2PPA**, **1PPA**, and **1MPA** based on the spectroscopic data. All energy values are taken from the onset of the emission spectra. It exemplifies the impact of reducing the conjugation length of the pyrimidine acceptor. While for **2PPA** and **1PPA**, the 3LE state localized on the 2-phenylpyrimidine (2PhPy) and 1-phenylpyrimidine (1PhPy) is the lowest triplet state with an onset at around 2.9 eV, this is no longer the case when the conjugation is

Table 1 Glass transition (T_g), recrystallization (T_r), melting (T_m), and decomposition at 5 wt% loss (T_d) temperatures determined from differential scanning calorimetry and thermogravimetric analysis. HOMO and LUMO values of **2PPA**, **1PPA**, and **1MPA**, calculated from cyclic voltammetry and the optical bandgap, respectively

	T_g (°C)	T_r (°C)	T_m (°C)	T_d (°C)	HOMO (eV)	LUMO (eV)
2PPA	108	127, 147	305	359	−5.12	−2.00
1PPA	150	229	304	417	−5.11	−1.71
1MPA	138	222	273	413	−5.10	−1.55



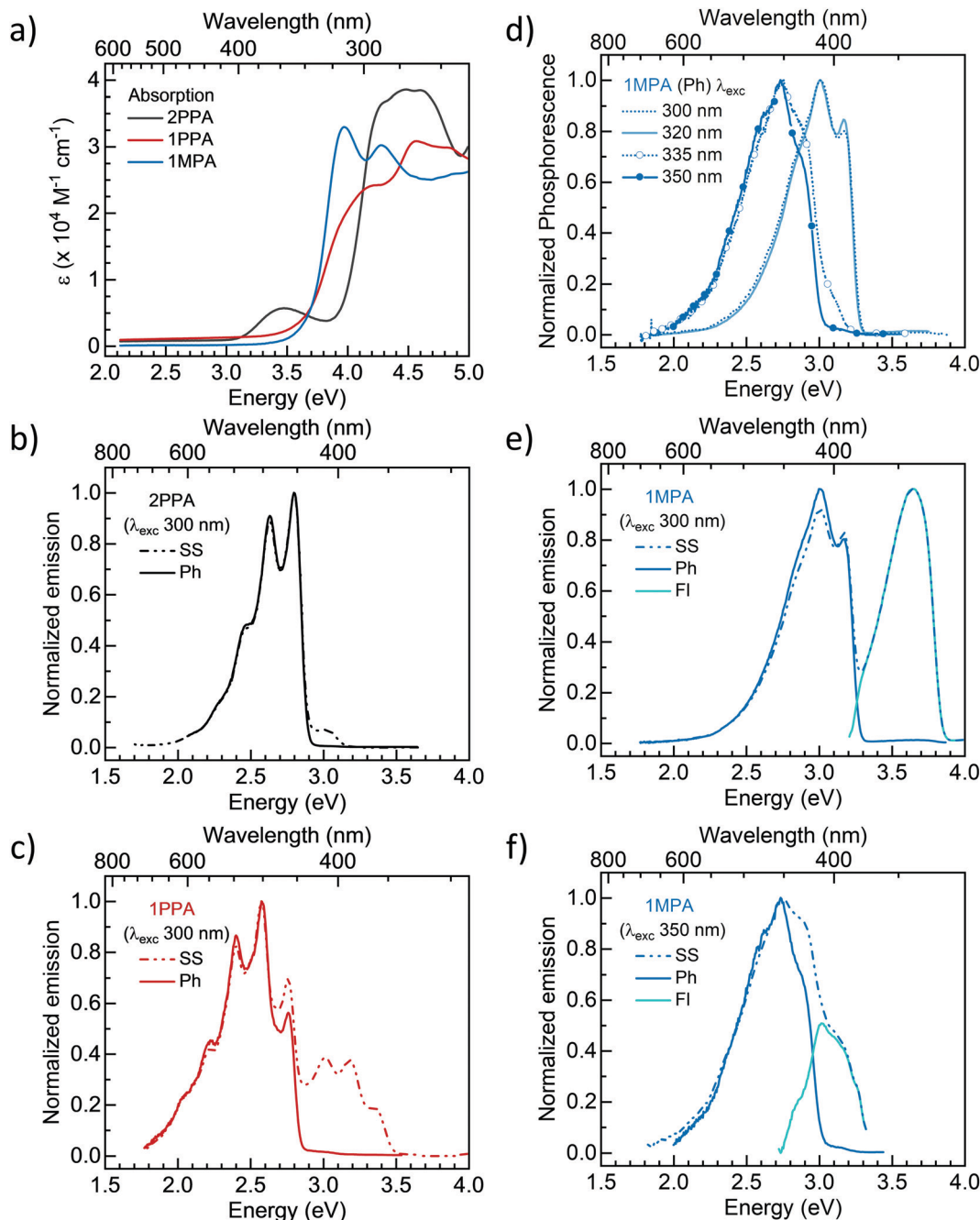


Fig. 3 (a) The absorption spectra of **2PPA**, **1PPA**, and **1MPA** in toluene solution at 300 K. (b) The emission spectra of **2PPA** taken at 77 K in a toluene glass under steady-state conditions (SS), so that both fluorescence and phosphorescence show up, and taken with a delay of 150 ms after excitation and a gate width of 50 ms so that only phosphorescence (Ph) is visible. Excitation was at 300 nm. (c) The emission spectra were taken under the same conditions for **1PPA**. (d) The phosphorescence (Ph) spectra of **1MPA** in toluene glass at 77 K, taken for different excitation wavelengths as indicated. (e) Comparison of the Ph and SS spectra for excitation at 300 nm and (f) at 350 nm. Also indicated is the fluorescence spectrum obtained from subtracting the Ph from the SS.

reduced in methylpyrimidine (MePy). The triplet state for the units pyrimidine and methylpyrimidine have 0-0 peaks at 3.50 eV and 3.42 eV, respectively (see Nishi *et al.*³⁶ and Fig. S5b, ESI[†]), *i.e.*, well above the onset of phosphorescence for the acridine moiety (Fig. S4 and S5b, ESI[†]). As a result, in **1MPA**, the lowest triplet state becomes one with a mixed CT-LE character, where the LE contribution is from the acridine unit. We also comment on the

unusual observation of an excitation wavelength-dependent phosphorescence. This implies that when exciting into the acridine-based singlet LE state, the rates of intersystem crossing to its triplet state and phosphorescence can compete with the charge transfer required to form the CT-based singlet or triplet state. Evidently, the twists between the units slow down the charge transfer, which needs to take place through a bond.



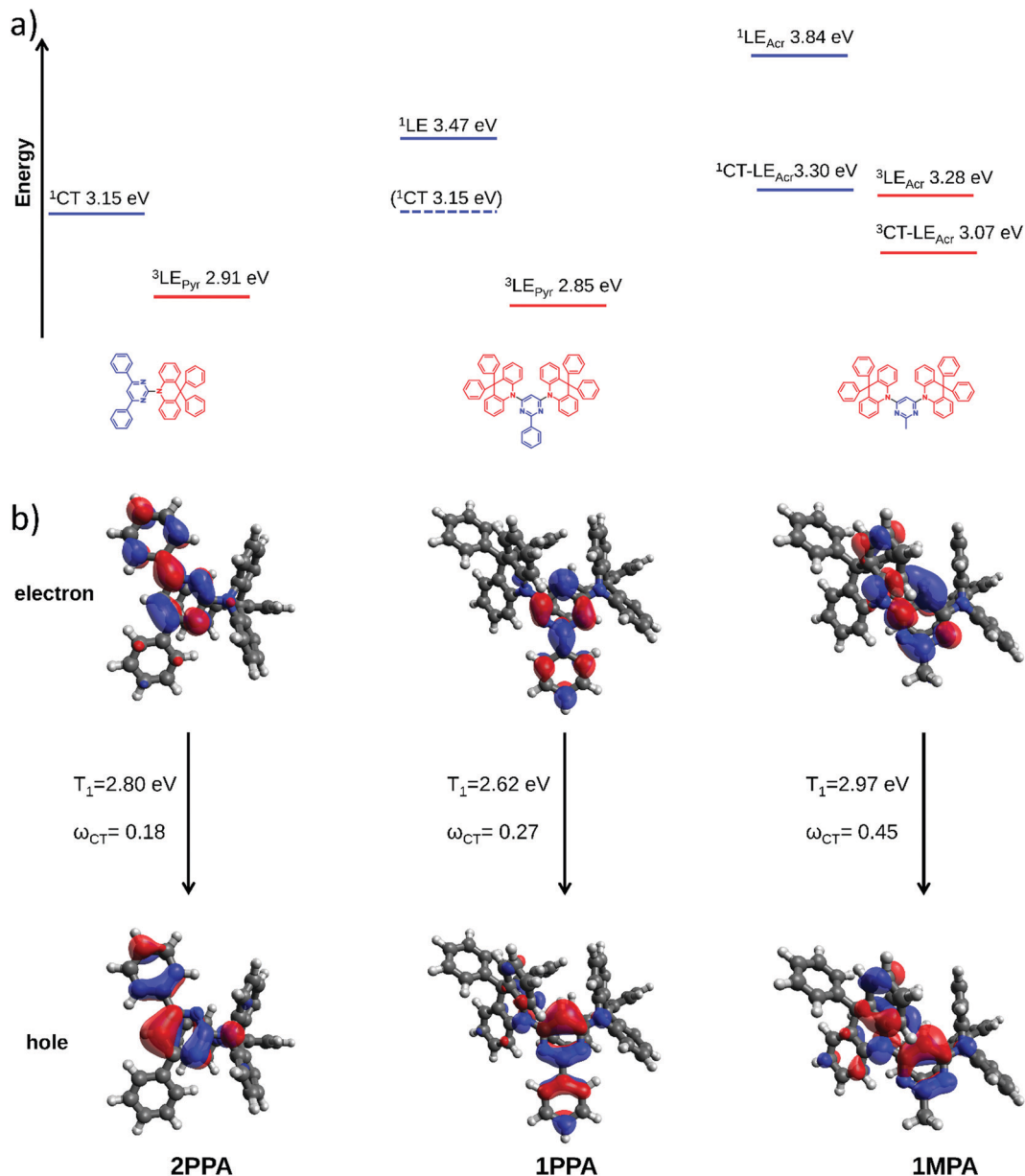


Fig. 4 (a) Excited state energy level diagram for **2PPA**, **1PPA**, and **1MPA**, based on the spectroscopic data, along with their structure (donor in red and acceptor in blue). The values are taken from the onset of the emission spectra. (b) TDA-DFT computed natural transition orbitals, emission energies, and charge transfer numbers for the lowest triplet state for **2PPA**, **1PPA**, and **1MPA** at the M06-2X/6-31+G(d,p) level. Pyr represents pyrimidine acceptors and Acr the acridine donor. A side view is displayed in Fig. S8 (ESI[†]).

Theoretical calculations

For a deeper understanding of the excited states, we turn our attention to the electronic structure calculations. We find that for all molecules, acridine units lose planarity, and their central ring adopts a distorted boat conformation both at the ground and excited state geometries. This boat conformation is in agreement with results on related compounds.^{37,38} At the same time, there is an intramolecular pi-stacking arrangement between the phenyl ring of one of the acridine donor units and the pyrimidine acceptor moiety (Fig. S7, ESI[†]). The obtained conformations suggest a strong electronic mixing between the donor and acceptor units, consistent with the

observations made in the absorption spectra. The TDA-DFT calculated lowest excited singlet and triplet energies are in very good agreement with the fluorescence and phosphorescence spectroscopic data, respectively, as summarized in Table 2. Calculations support the above spectroscopic assignment on the nature of the emissive triplet states. Fig. 4b illustrates the spatial extent of the hole–electron NTO pairs for the optimized lowest triplet state of **2PPA**, **1PPA**, and **1MPA**, along with the calculated charge transfer numbers ω_{CT} . These transitions can be described by a single NTO pair. The calculated relaxed T_1 state for **2PPA** and **1PPA** has a LE character and is localized on the pyrimidine unit, whereas the relaxed T_1 state for **1MPA** has



Table 2 TDA-DFT computed singlet ($S_1 \rightarrow S_0$) and triplet ($T_1 \rightarrow S_0$) emission energies of **2PPA**, **1PPA**, and **1MPA** at the M06-2X/6-31+G(d,p) level along with the energy values of fluorescence (Fl) and phosphorescence (Ph) taken from the onset of the spectra at 77 K in toluene solution for 300 nm excitation

Compound	Theory	Experiment	Theory	Experiment
	S_1 (eV)	S_1 (eV)	T_1 (eV)	T_1 (eV)
2PPA	3.23	3.15	2.80	2.91
1PPA	3.41	3.47	2.62	2.85
1MPA	3.50 ($S_2 \rightarrow S_0$)	3.84	2.97	3.28
	3.13 ($S_1 \rightarrow S_0$)	3.30 ^a		3.07 ^a

^a Excitation at 350 nm.

a mixed CT-LE character, with the LE part on the acridine. An energy level diagram based on the spectroscopic data is depicted in Fig. 4a. The NTOs for fluorescence can be found in Fig. S7 (ESI[†]). The emissive singlet for **2PPA** is localized on the pyrimidine, which is in contrast to the spectroscopic assignment of a CT state. This comes from the fact that the obtained gas-phase geometry from the TDA-DFT calculations leads to a distorted acridine. However, in the solid state, intermolecular interactions might favor pi-pi stacking configurations between the acridine units of the molecules and result in a twisted intramolecular CT state. For **1PPA**, the obtained lowest singlet is at 3.41 eV and has a CT character. The lowest singlet state for **1MPA** is at 3.13 eV, which displays a strong charge transfer character as depicted by the hole-electron pair natural transition orbitals and the electron-hole correlation plot of the Ω matrix (Fig. S9, ESI[†]) and has a weak oscillator strength $f = 0.0037$. Following optimization of the second lowest singlet state, S_2 , we find that it is located at 3.50 eV and has an oscillator strength $f = 0.0134$ (Fig. S7, ESI[†]). Although the charge transfer numbers obtained are similar for the two states, the electron-hole distance index Δr is larger for the S_1 state ($\Delta r = 3.26$ Å) than for the S_2 state ($\Delta r = 2.07$ Å), indicating a larger CT length for the S_1 state. The hole NTO for S_2 is delocalized across both acridine units. This could explain the experimentally observed excitation wavelength dependent emission described in the previous section. Given the relatively large energy difference between the S_2 and S_1 states and the fact that relaxation from the Franck-Condon state requires large conformational rearrangement, vibrational relaxation and internal conversion to S_1 might compete with radiative emission from the S_2 state leading to non-Kasha emission.³⁹

OLED properties

After having established that the acridine-pyrimidine based compound **1MPA** is the one with the highest triplet energy (> 3.0 eV) among the studied compounds, we employed it as a host for the fabrication of blue OLED. We combined it with the previously reported blue TADF emitters **mPTC**⁴⁰ and **OBA-O**⁴¹ as dopants (Fig. S10e, ESI[†]). In the case of **mPTC**, a doping concentration of 12 wt% was used based on photoluminescence quantum yield measurements (Table S1, ESI[†]). For the emitter **OBA-O** we used several doping concentrations (5, 8, and 13 wt%) with a view to examine the emission wavelength tunability of the

device. The choice of **mPTC** and **OBA-O** as TADF emitters was based on (i) their shallow HOMO levels (-5.12 and -5.15 eV, respectively), which were similar to that of **1MPA**, (ii) their triplet energies being lower than that of the host to allow for triplet exciton confinement and (iii) their high photoluminescence quantum yields. OLEDs were fabricated using the following simple device architecture: ITO (100 nm)/TAPC (30 nm)/EML (x wt% **mPTC** and **OBA-O** in **1MPA**, 30 nm)/TmPyPB (40 nm)/LiF (0.8 nm)/Al (100 nm). Here, ITO was used as a transparent anode, (1,1-bis[(di-4-tolylamino)phenyl]cyclohexane) TAPC and (1,3,5-tri(*m*-pyridin-3-ylphenyl)benzene) TmPyPB for hole and electron injection/transport, respectively, LiF/Al as the reflective cathode, whereas x denotes the doping concentration (in weight percentage) of the particular TADF emitter in the emissive layer (EML).

The configuration and the energy level diagram of the device with the EML (12 wt% **mPTC** in **1MPA**) are displayed in Fig. 5a. The main device characteristics are provided in Fig. 5b-d and Table 3. The OLED demonstrated a low turn-on voltage (V_{on}) of 3.25 V, greenish-blue electroluminescence at 491 nm with the full width at half maximum (FWHM) of 75 nm resulting in Commission Internationale de L'Eclairage (CIE) coordinates of (0.2, 0.4). The maximum external quantum efficiency (EQE_{max}) was determined to be 13.6%, which is in good agreement with PLQY of **mPTC** in **1MPA** (PLQY $\approx 70\%$, see Table S1 of the ESI[†])⁴⁰ if typical device outcoupling efficiency of about 20% is taken into account.⁴² The fabricated device exhibited low efficiency roll-off at practically useful brightness, *i.e.*, EQE dropped down to 12.3% at the brightness of 100 cd m⁻² and to 11.6% at 1000 cd m⁻² (Fig. 5b). Notably, EQE remained above 10% up to the very high brightness of 5000 cd m⁻². This behavior could be explained by the optimal device architecture and well-balanced electron and hole currents. The maximum luminance achieved by the device was nearly 44 000 cd m⁻². The properties of analogous devices based on the same host, yet the different TADF emitter **OBA-O** at various doping concentrations in the EML, are presented in Fig. S10 (ESI[†]) and Table 3. The **OBA-O** device with a similar doping concentration (13 wt%) as the previous one based on **mPTC** expressed comparable emission properties. Explicitly, peak emission wavelength, bandwidth, CIE color coordinates, and maximum EQE were found to be alike. On the other hand, the much steeper $I-V$ and $L-V$ curves in the **OBA-O** device indicated a significantly improved charge carrier mobility in the EML. Importantly, the reduced **OBA-O** concentration down to 5 wt% in the EML shifted the emission wavelength from greenish-blue (493 nm) to deep-blue (461 nm), demonstrating the potential of **1MPA** to host deep-blue TADF emitters. The picture of the deep-blue emitting OLED is shown in the inset of Fig. S10 (ESI[†]). The reduced emitter concentration concomitantly caused a broadening of emission bandwidth (up to 94 nm) and an accelerated EQE roll-off. In OLEDs, EQE roll-off typically arises from exciton-polaron quenching and from exciton-exciton annihilation. The increased roll-off with lower emitter concentrations can arise from either effect. Since there is a barrier of about 1.15 eV for electron injection onto the host yet not for injection onto the emitter, the likely scenario is that electron transport



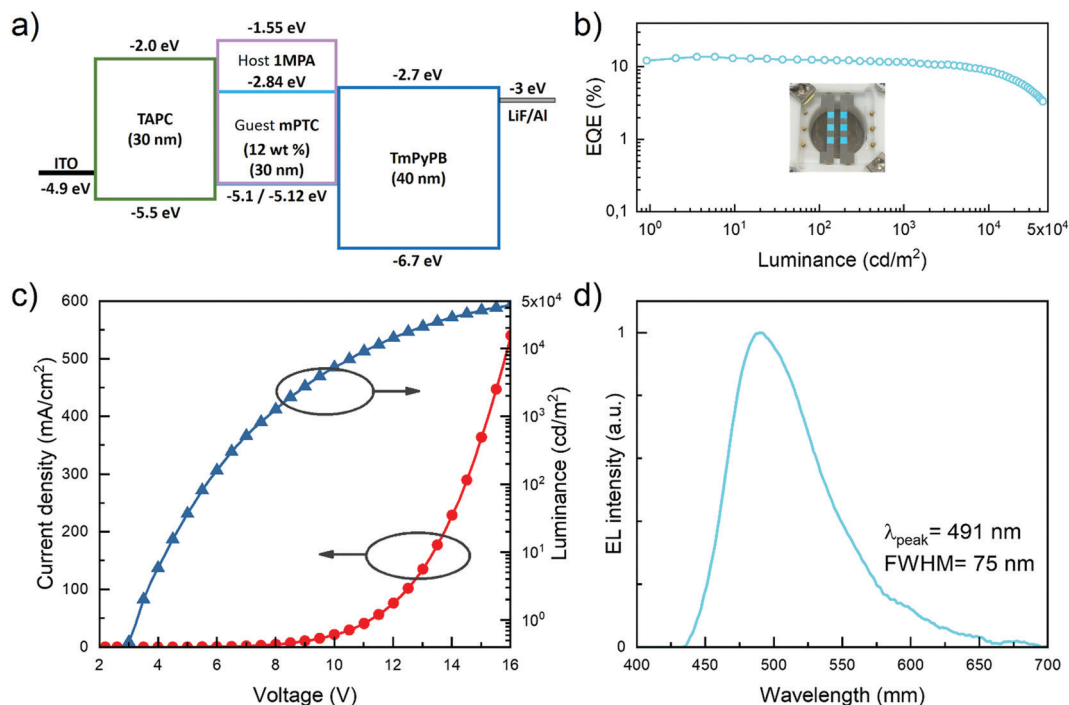


Fig. 5 Main properties of the TADF OLED based on the **1MPA** host doped with 12 wt% of **mPTC**. (a) Energy level diagram. (b) EQE vs. luminance. (c) Current density and luminance vs. applied voltage. (d) Electroluminescence spectrum with working device picture shown in the inset.

Table 3 Main parameters of blue TADF OLEDs based on the **1MPA** host doped with **mPTC** and **OBA-O** emitters

Emitter	V_{on}^a (V)	EQE ^b (%)	L_{max}^c (cd m ⁻²)	CE_{max}^d (cd A ⁻¹)	LE_{max}^e (lm W ⁻¹)	λ_{max}^f (nm)	FWHM ^g (nm)	CIE 1931 ^h (x, y)
mPTC (12 wt%)	3.25	13.6/12.3/11.5	43 911	33.3	27.5	491	75	(0.2, 0.4)
OBA-O (13 wt%)	2.75	14.2/11.9/8.9	30 247	36.6	36.4	493	83	(0.2, 0.38)
OBA-O (8 wt%)	3.25	14.9/6.8/2.9	8449	26.2	24.5	475	93	(0.16, 0.21)
OBA-O (5 wt%)	3.5	10.6/3.5/1.6	52 83	14.4	12.1	461	94	(0.16, 0.18)

^a Turn-on voltage at 1 cd m⁻². ^b Maximum EQE/EQE at 100 cd m⁻²/EQE at 1000 cd m⁻². ^c Maximum brightness. ^d Maximum current efficiency. ^e Maximum luminous efficiency. ^f Peak emission wavelength. ^g Full width at half maximum. ^h Commission Internationale de L'Eclairage color coordinates (x, y).

proceeds by hopping between emitter molecules while hole transport occurs *via* the host, where hole injection is barrier-free. This results in a charge carrier imbalance that favors exciton-hole quenching. However, at low emitter concentrations percolative transport across emitting sites is no longer possible, so that the recombination zone narrows towards the electron injecting side. This decrease in recombination zone implies an increase of both exciton and charge carrier density, so that in addition to exciton-hole quenching there can also be a contribution from triplet-triplet annihilation. (see Fig. S10b, ESI[†]). We note that the dominant lifetime of delayed fluorescence in **mPTC** (2 μ s) is shorter than that in **OBA-O** (9 μ s), consistent with its lower susceptibility to exciton quenching processes (Fig. S12 and Table S2, ESI[†]). Furthermore, we fabricated an identical **mPTC**-based device where the **1MPA** host is replaced with the benchmark host **mCBP**. In Fig. S11 (ESI[†]) we provide the comparison of OLED properties of the two identical devices differing only by the host material (**1MPA** vs. benchmark **mCBP**). Due to the lower LUMO of **mCBP** compared to **1MPA**, many more electrons are being injected into the **mCBP** host

resulting in significantly higher current densities flowing through the device, and subsequently, higher luminance attained at the same driving voltage. EL spectra of both devices utilizing the same **mPTC** dopant concentration (12 wt%) are found to be very similar. The device featuring benchmark **mCBP** host exhibits a higher EQEmax of 21% and similar efficiency roll-off as the other one based on the **1MPA** host. Nevertheless, the obtained results show that high-triplet-energy host **1MPA** is suitable for the fabrication of efficient blue TADF OLEDs, delivering low efficiency roll-off even beyond the practical brightness of 1000 cd m⁻².

Conclusions

In this work, we present three newly synthesized host molecules that adopt acridine and pyrimidine units. The bulky acridine provides superior thermal and morphological properties, further improved when two acridines are used, like in **1PPA** and **1MPA**. The donor and acceptor combination allows



transporting both holes and electrons. Throughout the series, it is possible to observe a decrease in the conjugation of the acceptor moiety within the hosts. When the acceptor unit is limited to only one pyrimidine ring (**1MPA** host), the triplet energy is the highest, with a value of 3.07 eV. Through theoretical calculations and spectroscopy, we attribute its nature to a mixed CT–LE state. Finally, to demonstrate the potential of these molecules, **1MPA** is employed as the host material to fabricate blue TADF OLEDs. Using the emitter **mPFC** we achieved a sky-blue OLED with electroluminescence at 491 nm and a maximum EQE of 13.6%, combined with a low roll-off. The host **1MPA** is also shown to be suitable for deep-blue emitters like OBA-O, delivering blue OLEDs with color coordinates of $x = 0.16$ and $y = 0.18$.

Author contributions

F. Rodella performed the synthesis and characterization of the host materials, as well as the thermal and electrochemical analysis. R. Saxena and S. Bagnich performed the photo-physical studies. D. Banevičius and G. Kreiza fabricated OLED devices. S. Athanasopoulos was responsible for DFT calculations. S. Juršėnas, K. Kazlauskas, A. Köhler and P. Strohriegel supervised the experiments and corrected the manuscript.

Conflicts of interest

There are no conflicts to declare.

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